

ADVANCES IN THE UNDERSTANDING OF U₃Si₂-AI DISPERSION FUEL IRRADIATION BEHAVIOR

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SWELLING BEHAVIOR DURING IRRADIATION

- Around 1987, at the beginning of ANL's participation in the ANS Project, we had found that certain intermetallic compounds, e.g., UAl_x, U₃Si₂, and USi, exhibited very stable swelling behavior to full burnup with LEU, while other compounds, e.g., U₆Fe, U₆Mn, and U₃Si became unstable before full burnup was achieved.
- We concluded that the unstable, or break-away, swelling is associated with irradiation-induced amorphization and that the compounds exhibiting stable swelling remained crystalline during irradiation.

- Irradiation of MEU and HEU U₃Si₂ showed that the swelling fell below the LEU values. Detailed analysis revealed a two-stage swelling behavior with an apparent fission rate effect.
- The fission gas bubbles in U₃Si₂ were distributed in a very regular pattern to the highest fission densities reached in our tests in the ORR (for ~65% burnup in HEU particles). The bubbles grew in diameter with increasing fission density but did not coalesce.



Fig. 1. Swelling of various LEU intermetallic fuel compounds as a function of accumulated fissions in fuel.







Bubble morphology in HEU, U_3Si_2 @ ~70% Bu, ORR (SEM)

- Further investigation revealed two problems:
 - Neutron diffraction studies of fuel irradiated in the IPNS at ANL indicated that both U_3 Si and U_3 Si₂ became amorphous at very low doses.
 - Our swelling model could not reproduce the bubble distribution seen in U_3Si_2 .

- The solution was to postulate the formation of subgrains in U₃Si₂ at the fission density of the knee of the swelling curve. These could be formed by subdivision of grains of crystalline material or by recrystallization of amorphized material.
 - Subgrains had been found in UO_2 and U_4O_9 , with similarities to U_3Si_2 in the fission gas bubble morphology.
 - Ion simulation experiments gave evidence of grain refinement.
 - We have now found evidence of subgrains in U₃Si₂ irradiated in the HFIR.

- ANS irradiation conditions of temperature, fission rate, and fission density were well beyond those in the ORR irradiations.
- Examination of samples irradiated in the HFIR target region has provided further insights into the behavior of U₃Si₂.
- We postulate a multi-stage process:
 - The outer portion of the U₃Si₂ particle reacts with AI to form a compound with a UAI₃-type structure, U(SiAI)₃. This material has the high stability characteristic of UAI₃; no fission gas bubbles are seen.

- Inside this zone subgrains form in the unreacted U₃Si₂ when the fission density reaches the knee of the swelling curve, and typical swelling ensues.
- The center of many fuel particles, however, show bubbles characteristic of the unstable, amorphous materials. We postulate that depletion of the U has led to the formation of an unstable phase in high-burnup HEU fuel particles. We think that this phase may be the peritectoid USi₂, analogous to U₃Si. Break-away swelling is restrained, however, by the outer U(SiAl)₃ shell.







Bubble morphology and Al interaction @ ~85% Bu, 250°C, HFIR

THERMAL CONDUCTIVITY CALCULATIONS

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THERMAL CONDUCTIVITY MODEL

- Model developed and implemented in the Dispersion Analysis Research Tool (DART) code by J. Rest, ANL.
- Thermal conductivity is a function of fuel and porosity volume fractions, which are calculated based on a mechanistic model of fuel particle swelling.
 - Fuel particle swelling model has evolved. Calculations now based on multiphase model of fuel swelling which fits both ORR and HFIR data.

Model Basis

- Spherical fuel particles are mixed into matrix aluminum.
- As-fabricated porosity is mixed into fuel meat.
- Irradiation-induced porosity is mixed into fuel particles.
- Aluminum matrix is assumed to be the continuous medium.
- Successive application of classical mixing formula gives equation for thermal conductivity as a function of fuel and porosity fractions.
- Fuel meat volume is assumed to remain constant as fuel particles swell--porosity is first closed then aluminum removed from meat.

Thermal Conductivity Equation

$$k_e^m = k_{al} \left[z_1 + z_2 F_v^{2/3} + z_3 \left(k_e^g / k_{al} \right) F_v^{1/3} \left(1 - F_p^{2/3} \right) + z_4 F_p^{2/3} + z_5 \left(F_v F_p \right)^{2/3} \right]$$

where

 k_e^m , k_{al} , and k_e^g are the thermal conductivity of the fuel meat, matrix aluminum, and U₃Si₂ particles containing irradiation-induced porosity; F_v and F_p are the fuel and as-fabricated porosity volume fractions; and

 $z_1 - z_5$ are geometric constants.

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• The physical limits:

$$\circ \quad k_e^m = k_{al} \quad \text{for} \quad F_p = 0, F_v = 0$$

 $\circ \quad k_{em}=0 \quad \text{for} \quad F_p=1$

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$$k_{em} = k_{al}$$
 for $F_v = 1$, $k_{eg} = k_{al}$

$$\circ \quad k_{em} = 0 \quad \text{for} \quad F_v = 1, \ k_{eg} = 0$$

imply that $z_1 = +1$, $z_2 = -1$, $z_3 = +1$, $z_4 = -1$

 z₅ is determined by fit to thermal conductivity data for unirradiated fuel to correct for idealized geometry assumption:

$$\circ$$
 z₅ = -0.3275

Model Limitations

- Treats only single size of fuel particle.
- Invalid when aluminum matrix is discontinuous, i.e., for less than 35 to 40 vol% aluminum.
- Does not account for any degradation of aluminum thermal conductivity due to irradiation.
- Accuracy of the results depends on the accuracy of the thermal conductivity and fuel swelling models and data. Fuel swelling depends on fission rate, total fissions, and temperature.



MULTINODE DART CALCULATIONS



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